

Synthesis and Characterization of BaTiO₃ Nano particles by Organic Precursor Method

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Abstract

BaTiO₃ nano powders prepared by using organic acid method. IR spectrum and EDS were used to identify the composition of the nano particles. XRD spectrum indicated the polycrystalline nature of the particles. Surface morphology of the particle was studied by scanning electron microscope (SEM). No pits and pin holes were found in the surface.

Keywords: BaTiO₃, EDS, XRD, SEM, etc.,

1. INTRODUCTION

Ferroelectric thin films are very promising for a wide range of application such as high dielectric capacitors, insulating surface layer, non volatile memories with low switching voltage, infrared sensors and electro optic devices. BaTiO₃ family ceramic thin films are receiving extensive investigation as the charge- storage dielectric for G-bit dynamic random access memory(DRAM), on-chip components such as de-coupling capacitors and positive temperature coefficient of resistance (PTCR) thermistors due to high dielectric constant, low dielectric loss, low leakage current, low temperature coefficient of dielectric constant and thermal stability[1].

Synthesis of BaTiO₃ nanopowders and thin films has attracted great attention over the last few decades due to the desirable properties and applications. BaTiO₃ particles could be produced by employing various chemical methods such as sol-gel technique[2], metal-organic processing [4], co-precipitation[3], hydrothermal treatment [5,6] and mechanochemical synthesis[7]. Organic acid precursor is a promising method that offer relative low cost, uniform size, homogeneous powder and high purity of the ceramics.

This films of BaTiO₃ could be prepared by different technique like r.f sputtering [8], pulsed laser ablation [9] and metal- organic chemical vapour deposition [10] etc. Despite that several technique have been explored to deposit thin film of BaTiO₃ less attention has been devoted to thermal evaporation. In the present study, we have reported about the BaTiO₃ nanopowder prepared by using organic acid precursor method using titanium dioxide TiO₂ as a source of titanium and oxalic acid as organic acid..

2. EXPERIMENTAL

2.1 Synthesis of BaTiO₃ nanopowder

BaTiO₃ nanopowders were synthesized using organic acid precursor method. The starting materials used were barium chloride BaCl₂·2H₂O, TiO₂ powder and Oxalic acid. A solution of Ba: Ti : Oxalic acid mole ratio 1: 1: 1 was stirred and evaporated at 70°C till a clear, viscous resin was obtained, then dried at 110°C for 20 hours. The precursor formed was heated at 900°C for 2 hours to form BaTiO₃ nanopowder.

2.2 Characteristics of BaTiO₃ nanopowder

FTIR and Energy Dispersive X-ray micro analyzer (EDX) were used for the identification of the chemical composition. The XRD patterns of the resulting products were obtained from X- ray powder diffraction with CuK α radiation. The micrograph of BaTiO₃ was examined by direct observation via scanning electron microscope (SEM).

3. RESULT AND DISCUSSION

Fig.1 shows infrared spectra of the prepared BaTiO₃ nanopowder. A strong absorption feature at 3205cm⁻¹ has been identified as the stretching mode of OH ions interpreted in the film and is commonly found in pervoskite films grown at low temperature by more conventional techniques.

This absorption referred to as hydroxyl defects because of the high mobility of hydrogen in pervoskite materials and is usually annealed out at temperatures near the phase transformation temperatures. The peaks at 809 cm⁻¹, 1290 cm⁻¹, 1663 cm⁻¹, 2357 cm⁻¹, 2521cm⁻¹ and 2646 cm⁻¹ are characteristics of the CO₃²⁻ group. The band at 3205 cm⁻¹ assigned to bulk hydrogen impurities [11].

It is well known that hydrogen impurities can penetrate several ternary oxides with perovskite structure as H^+ bonded to lattice oxygen in the form of an OH^- [12]. These protons can compensate for the cation charge defect due either to reduced centers such as Ti^{3+} or to cation vacancies in nonstoichiometric samples. The weak shoulder near 2900 cm^{-1} , 1958 cm^{-1} and 1042 cm^{-1} which are characterized by the absorption of CH_2 groups. The broad bands at 557 cm^{-1} and 400 cm^{-1} are due to Ti-O vibrations in $BaTiO_3$. FTIR analysis suggested that defects of negligible percentage are present in the lattice of $BaTiO_3$ nano powder.

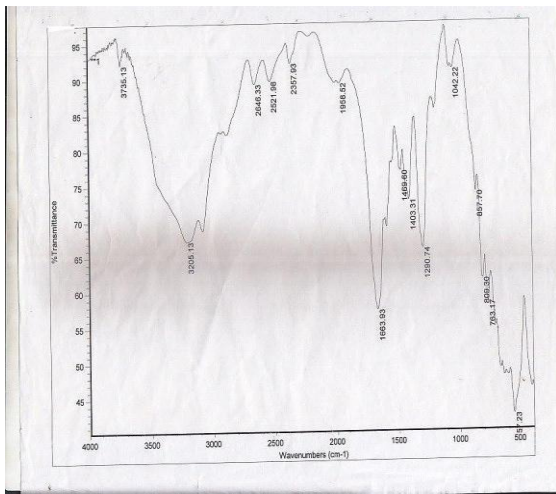


Fig. 1 FTIR spectrum of $BaTiO_3$ nano powder

Fig.2 shows the EDS spectrum of as prepared $BaTiO_3$ nano powder. Elemental composition analysis indicated the presence of Ba, Ti and O in the synthesized $BaTiO_3$ nano powder.

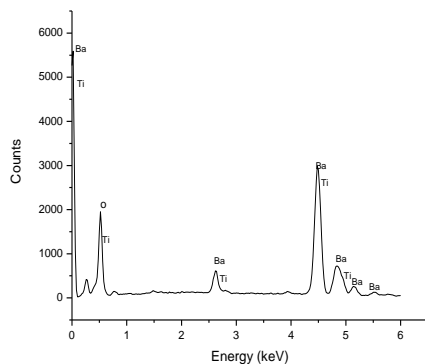


Fig.2 EDS spectrum of $BaTiO_3$ nano powder

A typical X-ray diffraction (XRD) pattern of $BaTiO_3$ powder as shown in the Fig.3.

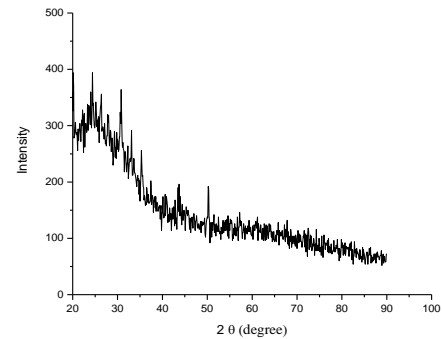


Fig. 3 XRD pattern of $BaTiO_3$ nano powder.

The crystallite size of $BaTiO_3$ for the most intense peak determines from the X-ray diffraction data using the Debye-Scherrer formula.

$$d_{RX} = kd/\beta \cos \theta$$

Where d_{RX} is the crystallite size to account for particle size $k = 0.9$ is a correction factor to account for particle shapes, F is the full width at half maximum (FWHM) of the most intense diffraction peak, λ is the wavelength of Cu target = 1.5406 \AA , and θ is the Bragg's angle. The small peaks are observed at $2\theta=24.4^\circ$ with a d value of 3.64 \AA , at $2\theta=30.712^\circ$ with a d value of 2.908 \AA , at $2\theta=35.34^\circ$ with a d value of 1.81 \AA . The peaks at $2\theta = 24.4^\circ$, 30.71° , 35.34° and 44° are respectively results from (001), (101), (110) and (002)/(200) planes of $BaTiO_3$. The resulting XRD and lattice parameters exactly match $BaTiO_3$ pattern (JCPDS 31-174).

A typical scanning electron micrographs of $BaTiO_3$ nano powder as shown on Fig. 4. The particulate phase in the micrograph is the $BaTiO_3$ grains and the voids seen in the micrograph. The micrograph indicates that the average grain size of $BaTiO_3$ nano powder varies from 90.68 nm to 133 nm .

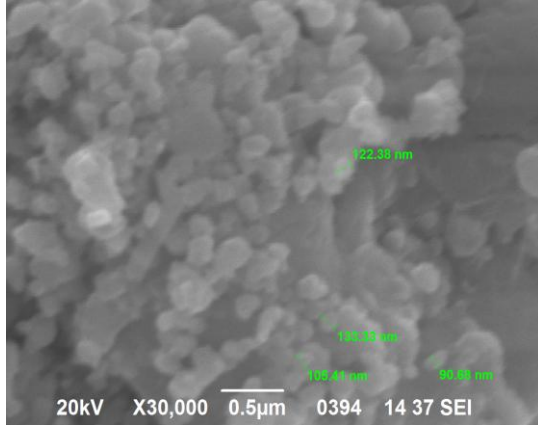


Fig.4 SEM micrograph of BaTiO₃ nano powder.

The grain size of vacuum evaporated BaTiO₃ thin films are relatively large compared to as prepared BaTiO₃ nano powder. Arlt.et.al [13] have reported that BaTiO₃ exists predominantly in the tetragonal structure for grain size of 1.5 µm and above and that, for an average grain size below 1.5 µm, tetragonal and orthorhombic phases coexist. Therefore, it seems that BaTiO₃ films prepared in this study also exists as a mixed system of tetragonal and orthorhombic or pseudocubic structure. Our X-ray studies also support this conclusion, because the peaks corresponding to the (002) and (200) planes are not well resolved.

4. CONCLUSION

We have successfully synthesized nano powder of BaTiO₃ by low cost organic acid precursor method using commercially available chemicals such as oxalic acid, TiO₂ and BaCl₂. The composition of BaTiO₃ nanopowder were confirmed by IR and EDX analysis. XRD indicated the predominantly amorphous nature of the film. The presence of small peaks indicated the presence of crystallites of very low dimension in the amorphous region. No pits, pin holes and dendritic features are found on the surface in SEM micrographs.

1. Scott, J.F., Annu. Rev. Mater. Sci., 28(1998)79.
2. Tangwiwat, S. and Milne, S.J., J. Non. Crystalline Solids, 351 (2005) 976.
3. Kirby, K.W., Mater. Res. Bull, 12 (1998) 881.
4. Shaikh, A.S. and Vest, G.M., J. Am. Ceram. Soci., 69 (1986) 682.
5. Boulos, M., Fritsch, S., Mathieu, F., Durand, B., Lebey, T., and Bley, V., Solid State Ionics, 176 (2005) 1301.
6. Tsumara T., Matsnoka K. and Toyoda M., J. Mater. Sci. Technol., 26(1) (2010) 33.
7. Stojanovic, B.D., Simowes, A.Z., Santos, C.O., Jovalekic, C., Mitic, V.V., and Varela, J.A., J. Euro. Ceram. Soci., 25(2005)1985.
8. Bhattachariya,P., Komeda,T., Park,D. and Nishioka,Y., Japan J. Appl. Phys., 32(1993)4103.
9. Yoon, S.G., Lee, J. and Safari, A., Integrated Ferroelectrics., 7(1995)329.
10. Tahan, D., Safari, A. and Klin, L.C., J. Am. Ceram. Soci., 79(1996)1593.
11. Kapphan, S. and Weber, G.R., Ferroelectrics, 37(1981) 673.
12. Baikov, Y.M. and Shalkona, E.K., J. Solid State Chem., 4(1994)1875.
13. Arlt, G., Hennings, D. and Dewith, G., J. Appl. Phys., 58(4)(1985)1619.